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# Insights into the intrinsic mechanisms underlying the ultra-highly efficient degradation of PFOA over S-scheme heterojunction of Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub>/BiOF



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#### ABSTRACT

Perfluorooctanoic acid (PFOA) is refractory owing to the strong C-F bonds and is extremely difficult to be degraded by traditional photocatalysts. In this study, an S-scheme heterojunction was successfully synthesized by incorporating BiOF with wide-bandgap  $Bi_7O_5F_{11}$ . Theoretical calculations, in situ X-ray photoelectron spectroscopy and Kelvin probe force microscopy verified that a build-in electric field was created in the heterojunction, facilitating the separation of electron-hole pairs. Thus, the photoexcited electrons in the conduction-band (CB) of BiOF rapidly recombined with holes in the valence-band (VB) of  $Bi_7O_5F_{11}$ , leading to enrichment of the highly reductive electrons in CB of  $Bi_7O_5F_{11}$  and strongly oxidative holes in VB of BiOF. Additionally, the strong interaction between  $Bi_7O_5F_{11}/BiOF$  and PFOA is conducive to electron transfer from catalyst surface to PFOA. Thus, BOF-2 displayed superbly-high photoactivity to decompose PFOA. PFOA (5 mg/L) was decomposed within 1 h, and 53% of total-organic-carbon was removed in 2 h of irradiation.

#### 1. Introduction

Per- and polyfluoroalkyl substances (PFAS) have been used for several decades in many industrial and commercial applications, thus leading to their ubiquitous occurrence and detection in the environment [1]. Perfluorooctanoic acid (PFOA, C<sub>7</sub>F<sub>15</sub>COOH) is a common PFAS of primary concern and is the dominant PFAS homolog in the environment due to earlier regulations on its sulphonate counterpart, perfluooctanesulfonic acid (PFOS) [2]. The strong electronegativity of F atoms and the strong carbon-fluorine bonds (bond energy 533 kJ·mol<sup>-1</sup>) engender PFOA extremely resistant to decompose, gaining it the moniker "the forever chemical" [3,4]. Additionally, PFOA is ionic and highly soluble, which makes aquatic environments as its major sink. Therefore, it is frequently detected in tap water (ng/L), natural water bodies ( $ng/L-\mu g/L$ ), and even in the blood of humans and wildlife [5–7]. A great number of studies documented that PFOA had various toxic effects, such as congenital disability, weakened immunity, growth retardation and so on [8,9]. In 2022, the United States Environmental Protection Agency (US EPA) established interim health advisory level of 0.02 ng/L for PFOS and 0.004 ng/L for PFOA in drinking water [10]. It is very challenging to develop highly efficient techniques to remove PFOA

from water to protect human health.

Several methods have been developed to remove or degrade PFOA in water, and among them photocatalysis is considered to be a prosperous solution considering its simple operation, high oxidation efficiency and environmental compatibility [11–13]. To ensure satisfactory photocatalytic performance to degrade PFOA, it is essential to design and fabricate photocatalysts with optimal band gap energy and band structure [11]. Among the many photocatalytic materials, bismuth oxyhalides (BiOX, X = F, Cl, Br, I) display attractive characteristics for PFOA degradation due to their unique two-dimensional layered structures and tunable energy-band properties [14,15]. In addition, the internal electric field (IEF) between halogen slabs and [Bi<sub>2</sub>O<sub>2</sub>] layers accelerate the charge migration and separation of electron-hole pairs, endorsing BiOX a good candidate for PFOA photocatalytic degradation [16].

Given the high energy C-F bonds in PFOA, it is generally considered that photocatalysts with broad energy-band would display better performance toward PFOA degradation [17]. BiOF, with a wide energy gap ( $E_g \approx 3.5$  eV), has a high redox capacity of electron hole pairs generated by photoexcitation, and therefore is an excellent candidate [18,19]. Although BiOF has a high redox capacity, it is fairly difficult for single-component photocatalysts to realize rapid separation of

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photogenerated carriers. Recently, construction of step scheme (S-scheme) heterojunctions, such as BiOBr/g-C<sub>3</sub>N<sub>4</sub> [20], Bi<sub>2</sub>Sn<sub>2</sub>O<sub>7</sub>/-BiOBr [21], TiO<sub>2</sub>/CsPbBr<sub>3</sub> [22], has been reported as a resultful strategy to overcome the aforementioned intrinsic shortcomings. In S-scheme heterojunctions, when two semiconductors with different Fermi levels (E<sub>f</sub>) come into contact, electrons would flow from the semiconductor with higher Ef to that with low Ef until they reach balance at the interface [23], constructing an internal electric field (IEF) and causing band bending at the interface. This construction would drive effective separation of photoproduced charges, preserving the strongly oxidative h+ and reductive e in the higher valence band (VB) and lower conduction band (CB), respectively. The detailed differences between S-scheme heterojunctions and the Z-scheme heterojunctions are summarized in Table S1 [24,25]. In our previous study [4], PFOA degradation was significantly improved by combining BiOF with the wide-band gap Metal-Organic Frameworks (MOFs) material. Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub> is a broadband gap semiconductor with outstanding performance, and it shares similar layered structure as BiOF [26–28]. It is anticipated that integrating them by in-situ synthesis might generate the most matching interface [29–31], which is more conducive to the flow of electrons and holes, and the internal electric field owned by Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub> and BiOF will also accelerate the electron transfer process [15].

Inspired by the above analyses, this study aims to develop a strategy for in situ synthesis of Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub>/BiOF photocatalysts by controlling the strong coordination between Bi(III) and NH<sub>4</sub>F and the reaction between Bi(III) and glycol to alkoxide. The regulation mechanism and principle of synthesis strategy are proposed and discussed. Meanwhile, the photocatalytic properties of the synthesized Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub>/BiOF samples to degrade PFOA were systematically studied. The S-scheme mechanism of Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub>/BiOF was discussed extensively according to theoretical calculation and in situ X-ray photoelectron spectroscopy (XPS) and in situ Kelvin probe force microscopy (KPFM). The dominant active species in the process of photoreaction were identified using trapping experiments and electron paramagnetic resonance (EPR) spectroscopy, and their attack ability was analyzed by theoretical calculation. The intermediate products of PFOA were identified by an ultra-performance liquid chromatography tandem mass spectrometry system (UPLC/MS/ MS). The mechanisms of photoresist and decomposition pathway of PFOA were proposed.

## 2. Experimental section

#### 2.1. Chemicals and reagents

Bi( $NO_3$ ) $_3$ ·5  $H_2O$  was provided by McLean Biochemical Technology Co., Ltd, Shanghai, China.  $NH_4F$  was purchased from Tianjin University Kowei Co., Ltd, Tianjin, China. Ethylene glycol (EG) was afforded by Tianjin Balens Biotechnology Co. Ltd., China. The ammonium oxalate (AO), KBrO $_3$  and benzoquinone (BQ) were provided by Guangfu Technology Development Co. Ltd., Tianjin, China. The t-butanol and 5,5-dimethyl-1-pyrroline N-oxide (DMPO) were purchased from Aladdin Industrial Corporation, Shanghai, China. Acetic acid was supplied by Fuchen Chemical Reagent Co., Ltd, Tianjin, China. PFOA with > 99% purity was supplied by Wellington Laboratories, Guelph, Canada. Ultrapure water prepared by a Deirect-Pure UP ultrapure water system was used for all solutions (Shanghai Lefeng Biotechnology Co., Ltd, China).

#### 2.2. Preparation of the photocatalysts

#### 1. Preparation of Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub>

 $Bi_7O_5F_{11}$  was prepared using a hydrothermal method: 2 mmol of NH<sub>4</sub>F and 1 mmol of Bi(NO<sub>3</sub>)<sub>3</sub>·5 H<sub>2</sub>O were respectively dissolved in 15 mL of mixed solution of acetic acid and water (the volume ratio of acetic acid and water was 1:5), and labeled as solution A and B. After ultrasonication for 30 min, solution A was dropwise added into solution B under constant stirring. The mixture was then stirred for

another 30 min, and sealed in a 100 mL polytetrafluoroethylene (PTFE) reactor, which was heated in an oven at 160 °C for 24 h. Subsequently, the prepared material was collected by centrifugation after cooling down to about 25 °C naturally, and washed several times with deionized water and ethanol. Finally, the cleaned white material was placed in an oven at 80 °C and dried overnight to obtain  $\rm Bi_{7}O_{5}F_{11}$  photocatalyst.

## 2. Preparation of $Bi_7O_5F_{11}/BiOF$

In a typical procedure, 2 mmol Bi(NO<sub>3</sub>)<sub>3</sub>·5 H<sub>2</sub>O was dissolved in 20 mL of ethylene glycol, and 2 mmol NH<sub>4</sub>F was dissolved in 20 mL of ethylene glycol/H2O mixed solutions, and the solutions were prepared under ultrasonication for 30 min. Next, the NH<sub>4</sub>F solution was dropwise added to the Bi(NO<sub>3</sub>)<sub>3</sub>·5 H<sub>2</sub>O solution, and the mixture was continuously stirred for another 30 min. The mixture was transferred to a 100 mL PTFE reactor, which was sealed and heated in an oven at 160 °C to react for 24 h. Thereafter, the solid products were collected by centrifugation and washed with deionized water and ethanol for several times. Finally, the greyish precipitates were placed in an oven at 80 °C to dry. For the prepared photocatalysts, when the solvent of NH<sub>4</sub>F was pure water, the product was labeled as BOF-1; while other photocatalysts were labeled as BOF-2, BOF-3, BOF-4, and BOF-5 according to the volume ratio of water/ethylene glycol in NH<sub>4</sub>F solvent (9:1, 7:3, 1:1, 3:7). The preparation process of Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub>/BiOF photocatalyst was presented in Scheme 1.

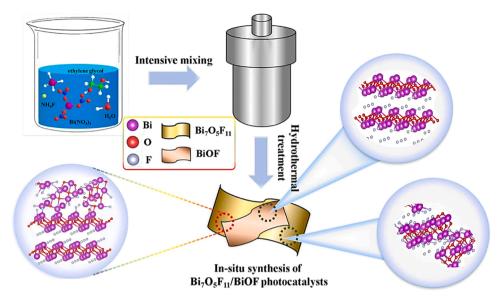
## 2.3. Characterization of the as-prepared photocatalysts

The crystal structures of the prepared Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub>/BiOF composites were analyzed employing an Ulitma IV X-ray diffractometer (XRD) by Cu-K $\alpha$  radiation under 40 kV and 100 mA with 8 $^{\circ}$ /min from 5 $^{\circ}$  to 80 $^{\circ}$ . The morphology and microstructure of Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub>/BiOF composites were inspected by JSM-7800 F field emission scanning electron microscopy (FESEM, Japan) and JEM-2800 high-resolution transmission electron microscopy (HRTEM, Japan). The chemical states, elemental compositions and charge transfer paths of the as-prepared photocatalysts were inspected by XPS and in situ XPS tests (XPS: ESCALAB250, Thermo Fisher Scientific, America). Kelvin probe force microscopy (KPFM) measurements were completed on a tap150E-G surface probe system (Bruker, Germany). The specific surface area and pore size distribution were determined by ASAP 2460 Brunauer-Emmett-Teller (BET, Micromeritics, America). Ultraviolet-visible diffuse reflection spectrophotometer (UV-vis DRS, U-3900 H, Techcomp, China) was used to measure the optical properties of Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub>/BiOF composites with BaSO<sub>4</sub> as reference, and the scanning range was 200-800 nm. A total organic carbon analyzer (TOC Multi N/C3100, Analytic Jena, Germany) was applied to determine the content of total organic carbon in the degradation process of PFOA. The details of photoelectrochemical measurement, degradation experiments and PFOA analysis are shown in the Supporting information (SI).

#### 3. Results and discussion

#### 3.1. Characterization of the as-prepared Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub>/BiOF

The XRD patterns of the as-prepared  $Bi_7O_5F_{11}$ /BiOF catalysts are shown in Fig. 1. The characteristic peaks of  $Bi_7O_5F_{11}$  were marked by blue diamond in the composite materials. The peaks at  $15.49^\circ$ ,  $22.31^\circ$ ,  $24.69^\circ$ ,  $26.67^\circ$ ,  $29.26^\circ$ ,  $31.27^\circ$ and  $34.88^\circ$  correspond to the  $(\overline{2}01)$ ,  $(\overline{2}02)$ , (202), (112), (401),  $(\overline{4}02)$ , (113) crystal facets of  $Bi_7O_5F_{11}$ , which are in agreement with those of pure  $Bi_7O_5F_{11}$  phase (JCPDS card No. 97–016–7074). The diffraction peaks belonging to BiOF were marked with purple plum blossom pattern, and the labeled peaks correspond to (001), (101), (110), (102), (112), (200) and (211) planes of the standard [JCPDS No. 97–020–1620]. The characteristic peaks of the two standard cards were provided at the bottom of Fig. 1. It can be seen that the



Scheme 1. Scheme of the preparation procedure of Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub>/BiOF.

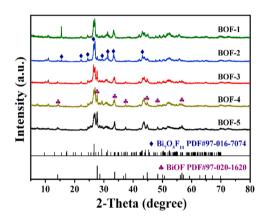


Fig. 1. XRD patterns of the as-prepared Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub>/BiOF.

diffraction peak intensities of the prepared materials (from BOF-1 to BOF-5) belonging to  $\rm Bi_7O_5F_{11}$  gradually weakened, while those of BiOF increased obviously, particularly for the diffraction peak intensity of (101) crystal plane at  $27.71^{\circ}$ . This indicates that the BiOF content in the composite materials gradually increased while that of  $\rm Bi_7O_5F_{11}$  decreased with the increase of the volume ratio of water/ethylene glycol in NH<sub>4</sub>F solvent.

In order to accurately analyze the relative contents of the two substances in the composite materials, the relative mass fractions of  $Bi_7O_5F_{11}$  and BiOF in the composites were estimated according to the following equation [32]:

$$w_i = \frac{I_i/RIR_i}{\sum_{i=1}^N I_i/RIR_i}$$

where  $W_i$  is the mass fraction of phase i (%),  $I_i$  is the strongest integral intensity of phase i,  $RIR_i$  value from MDI Jade 9.0. As shown in Table S2, the calculated mass ratios of  $Bi_7O_5F_{11}$  and BiOF in the composite materials of BOF-1 to BOF-5 were about 7:1, 3:1, 2:1 and 1:1. It is obvious that the mass ratio of  $Bi_7O_5F_{11}$  and BiOF in the composites could be well modulated by simply tuning the volume ratio of water and ethylene glycol in the NH<sub>4</sub>F precursor. It is worth noting that, when the volume ratio of water and ethylene glycol was 7:3 and 1:1, the mass ratio of  $Bi_7O_5F_{11}$  and BiOF in the prepared composites was basically 2:1. It was speculated that the precursor compositions might affect the crystal

nucleation and growth. The possible growth mechanism was speculated as follows: In pure water system, a portion of the Bi(III) ions are prone to interact with ethylene glycol to form alkoxide, while the remaining Bi (III) ions are hydrolyzed to form BiONO<sub>3</sub> [33,34]. In the presence of  $NH_4F$ , a Bi-ammonia complex  $(Bi[NH_3]_n^m)$  may be generated because of the strong coordination capacity of NH<sub>4</sub>F with Bi(III) ions [34]. With the increase of reaction temperature, the coordination complex could release Bi(III) ions gradually, which then reacted with F<sup>-</sup> to form BiF<sub>3</sub>, while the accumulation of local BiF3 will favor the formation of Bi7O5F11 [33]. However, with the increase of ethylene glycol content in the reaction system, the hydrolysis of Bi(III) ions would be suppressed, leading to the reduced accumulation of local BiF3, which is manifested by the decreased relative mass of Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub> in the composites. Meanwhile, due to the large viscosity and integration of ethylene glycol, the growth process of materials was also suppressed in the hydrothermal process as the ethylene glycol became dominant. Ethylene glycol existing in a long chain structure under the action of hydrogen bonding can also be used as the growth template of BiOX, and then BiOX forms microsphere structure through self-assembly process [35]. Therefore, it was speculated that when the volume ratio of water and ethylene glycol was adjusted from 7:3-1:1, ethylene glycol may be sacrificed as the growth template, and thus the mass ratio of BOF-4 was similar to that of BOF-3. In order to verify this hypothesis, the morphologies of the as-prepared materials were characterized by SEM and HRTEM.

As shown in Fig. 2, the microstructure of the photocatalysts varied distinctly due to the different preparation conditions. Fig. 2a illustrates that BOF-1 was mainly in an aggregated granular morphology, with the particle size of about 50 nm. However, the BOF-2 photocatalyst was dominated by sheet structure of about 100 nm. With the increase of ethylene glycol content in the system, the prepared BOF-4 to BOF-5 were in perfect flower-like microspheres, which were mainly stacked with regular hexagonal lamellar structure. From the lamellar edge structure described in the inset of Fig. 2c, it could be clearly observed that the hexagonal lamellar structure formed in BOF-3 was relatively loose. Compared with BOF-3, the hexagonal lamellae in BOF-4 were squeezed more tightly, and they were interspersed more closely.

The morphology analysis of BOF-3 and BOF-4 confirmed the hypothesis that more ethylene glycol in the precursor solution of BOF-4 was consumed as a growth template, thus reducing the inhibitory effects of ethylene glycol on the generation of  $\text{Bi}_7\text{O}_5\text{F}_{11}$ . The flower-like stacked structure of BOF-4 composite was obtained. On the other hand, BOF-5 was completely self-assembled into flower-like microsphere, and the edge of lamellar structure was "multi-layer step" shape.

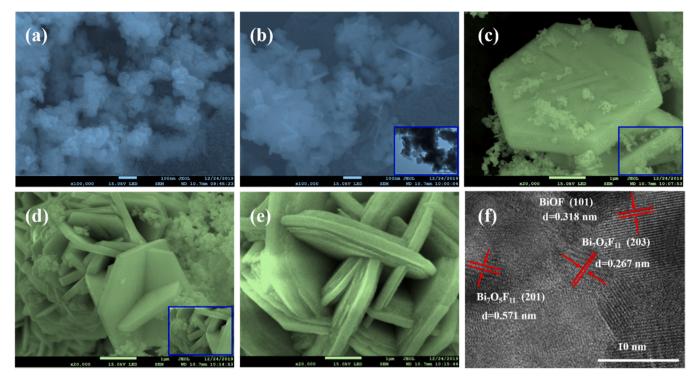


Fig. 2. SEM images of the prepared samples: (a) BOF-1, (b) BOF-2, (c) BOF-3, (d) BOF-4, (e) BOF-5; (f) HRTEM diagram of BOF-2.

As shown in Fig. 2f, the lattice spacings of 0.267 and 0.571 nm occurred on the edge corresponding to the (203) and (-201) facets of Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub>, while that at 0.318 nm belongs to the (101) crystal plane of BiOF, which also strongly confirms that the product is composed of Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub> and BiOF.

The optical properties of the photocatalysts were characterized by the UV-vis diffusion reflectance spectra. As can be seen from Fig. S2, the photo-responses of the prepared composite photocatalysts were mainly in the range of 200-300 nm, and the absorption edges were between 350 nm and 370 nm. It was confirmed that they were responsive to ultraviolet light, which was also consistent with the characteristics of the band structure of the composites. Therefore, Hg lamp was selected as the light source for subsequent photocatalytic degradation experiments. In order to further understand the influence of mass ratio of Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub> and BiOF on the energy band structure, the band gap energies (Eg) of the as prepared materials were calculated according to the equation of  $(\alpha h \nu)^{1/n}$ =  $A(h\nu - E_{\sigma})$  (n = 1/2 or 2) [36,37]. Since both Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub> and BiOF were direct band gap semiconductors, the value of n was 1/2 in the calculation process. The calculated band gap energies were as follows: BOF-1 < BOF-2 < BOF-3 < BOF-4 = BOF-5, which were 3.35 eV, 3.52 eV, 3.53 eV, 3.61 eV, respectively. The band gap energy of the composites shows an increasing trend with the decrease of the mass percentage of Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub> in the composites. A large band gap energy suggests that a higher excitation light energy is required, while the utilization rate of light energy decreases. However, the electron hole generated by these kinds of semiconductors equips with stronger redox ability [17].

The specific surface area, pore volume and pore size of the photocatalysts also directly affect their catalytic degradation activity. Large specific surface area can provide more reactive sites and facilitate the contact of contaminant molecules with the catalyst surface. The large pore volume and pore size may allow contaminant molecules to enter the pore structure, which facilitates the redox role of the photocatalysts. Therefore, the  $N_2$  adsorption characteristics of the prepared photocatalysts were compared and summarized in Table S3. Compared with other composites, BOF-2 had the largest specific surface area, pore volume and pore diameter, which were  $11.78 \text{ m}^2/\text{g}$ ,  $0.0970 \text{ cm}^3/\text{g}$  and

60.41 nm respectively. In particular, the pore size was much larger than that of other composites, which was favorable for adequate contact between the photocatalysts and pollutant molecules.

In order to investigate the chemical states and valences of each element in the prepared composite photocatalysts, XPS characterization was carried out. As shown in Fig. S3, the Bi 4  $f_{7/2}$  and 4  $f_{5/2}$  of all the materials were located at 159 eV and 164.3 eV, respectively, with an energy gap of 5.3 eV, which was in accordance with the valence state of Bi(III) in BOF [38]. As for O1s, two characteristic peaks were observed at 529.8 and 531.3 eV, which belong to lattice oxygen (Bi-O) and -OH absorbed on the surface (Fig. S3a). There were two binding peaks in the F 1 s spectrum (Fig. S3b). The peak at 679.9 eV was corresponding to Bi 4p, while that at 683.3 eV was mainly originated from F 1 s. The chemical bonds and existing chemical states of each element were similar in all the prepared materials although the mass ratio of  $Bi_7O_5F_{11}$  and BiOF was different. All the orbital peaks of the five composites were basically consistent without obvious shift change, which indicates the co-growth of  $Bi_7O_5F_{11}$  and BiOF during the synthesis process.

#### 3.2. Photocatalytic performance on PFOA degradation

The photocatalytic properties of the prepared composite photocatalysts were investigated by the degradation of PFOA under UV light irradiation. All photocatalysts did not show significant difference in the adsorption properties to PFOA during the 60 min adsorption process. In the absence of photocatalyst, the removal rate of PFOA was basically negligible after 2 h of UV irradiation. As shown in Fig. 3a, the asprepared BOF photocatalysts displayed more excellent photocatalytic performance compared with the commercial TiO2 (the removal rate of PFOA after 2 h illumination is about 23%). The order of degradation efficiency was as follows: BOF-2 > BOF-1 > BOF-3 > BOF-4 > BOF-5. Among them, BOF-2 exhibited the highest photo-oxidation activity, which degraded 15 mg/L PFOA almost completely in 1 h. Based on Langmuir-Hinshelwood kinetics model, the first-order kinetics of PFOA photodegradation with all the employed photocatalysts are displayed in Fig. S4. Apparently, the degradation rate of PFOA ( $k = 0.0459 \text{ min}^{-1}$ ) in the presence of BOF-2 was the highest, which was 2 and 41 times higher

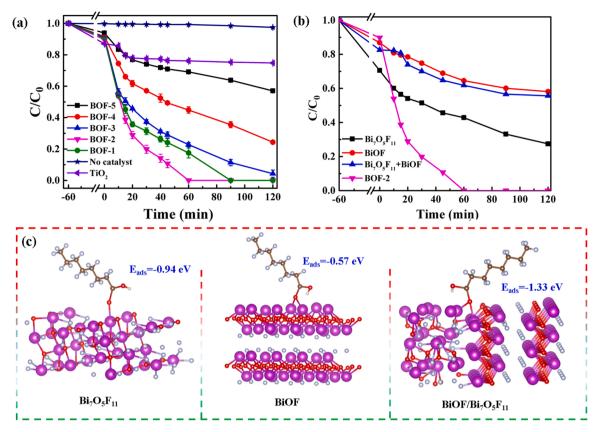


Fig. 3. (a) Photocatalytic properties of a series of photocatalysts ([PFOA] = 15 mg/L), (b) degradation performance curve of  $Bi_7O_5F_{11}$ , BiOF,  $Bi_7O_5F_{11} + BiOF$  and BOF-2 for PFOA; (c) DFT calculated adsorption energy of PFOA on  $Bi_7O_5F_{11}$ , BiOF and  $Bi_7O_5F_{11}$ /BiOF.

than that of BOF-1 ( $k=0.0223\,\mathrm{min}^{-1}$ ) and commercial TiO<sub>2</sub> ( $k=0.0011\,\mathrm{min}^{-1}$ ), respectively. Several kinds of photocatalysts have been fabricated for PFOA decomposition in previous studies, and Table S4 compared their efficiencies. It took much longer time (>3 h - 16 h) for these materials to photocatalytically degrade PFOA, but it took only 1 h for BOF-2 to remove 100% of PFOA. This suggests that BOF-2 exhibited ultra-high degradation performance to PFOA.

In order to verify that  $Bi_7O_5F_{11}$  and BiOF mainly exert photocatalytic degradation through the interface interaction, rather than the simple mixing effect of the two substances, the pure substances of  $Bi_7O_5F_{11}$  and BiOF were physically mixed in a mass ratio of 3:1 ( $Bi_7O_5F_{11} + BiOF$ ), and then reacted under the same degradation conditions. Indeed, as shown in Fig. 3b, BOF-2 exhibited much higher photocatalytic activity to degrade PFOA than those of  $Bi_7O_5F_{11} + BiOF$ , pure  $Bi_7O_5F_{11}$  and pure BiOF. This result magnifies that the intimate interfacial contact between  $Bi_7O_5F_{11}$  and BiOF plays a critical role in the improvement of photocatalytic performance, which is an effect of "1 + 1 > 2". Thus, BOF-2 was selected as the representative catalyst in the subsequent degradation experiment, and the optimum reaction condition was determined as 0.5 g/L of BOF-2, 40 mL of PFOA (15 mg/L) under 120 min irradiation (Fig. S5).

The interaction between contaminant molecules and photocatalyst surface plays a crucial part in photocatalytic reaction. To understand the role of the catalyst surface in the chemisorption process, DFT analysis was carried out to investigate the chemisorption interactions between PFOA and photocatalyst at molecular level. The adsorption energy  $(E_{ads})$  of PFOA on the  $Bi_7O_5F_{11}/BiOF\ (-1.33\ eV)$  was more negative than that on  $Bi_7O_5F_{11}$  and  $BiOF\ (-0.94\ eV\ and\ -0.57\ eV,\ Fig.\ 3c)$  respectively, indicating that there was a much stronger interaction between the BOF-2 surface and PFOA, thus favoring the electron transfer from the surface of the  $Bi_7O_5F_{11}/BiOF\ to\ PFOA$ , and the following degradation.

The reusability of BOF-2 during the photocatalytic degradation of

PFOA was tested. As shown in Fig. S7, the degradation rate of PFOA during the second and third cycles remained around 100%. After the recycle experiments, the BOF-2 was gathered and subjected to XRD analysis (Fig. S7b). The results indicated that the peak positions were well coincided with the original ones, demonstrating that it was very stable and resisted light erosion.

#### 3.3. Photocatalytic mechanisms

In order to explore the roles of various active species involved in the photocatalytic degradation of PFOA by BOF-2, a couple of related scavengers were applied, including ammonium oxalate (AO) as scavengers for the photogenerated holes (h<sup>+</sup>), KBrO<sub>3</sub>, tertiary butanol (TBA) and p-benzoquinone (BQ) to trap photogenerated electrons (e-), hydroxyl radicals ( $\bullet$ OH) and superoxide radicals ( $\bullet$ O $_2$ ), respectively [11]. As shown in Fig. 4a, the photocatalytic activity of BOF-2 was greatly suppressed by AO and BQ, and partly suppressed by KBrO<sub>3</sub>. However, the effect of TBA was negligible. This manifested that the e-, h+ and •O<sub>2</sub> played vital roles in PFOA decomposition by BOF-2. The results of the spin-trapping EPR measurement were consistent with those of the above experiments. TEMPO (2,2,6,6-tetramethylpiperidine-1-oxyl) was employed to monitor the generation of e and h, because it is consumed by e and h in carbonic acid and acetonitrile solutions, respectively, resulting in the decreased 1:1:1 characteristic signal of TEMPO [39]. As shown in Fig. 4(b-d), no corresponding free radical was detected in the three free radical detection systems without illumination. The signal intensities of e and h gradually weakened due to consumption of TEMPO, while the signal peak intensity of  $\bullet O_2^-$  increased with the extension of irradiation, indicating that e, h and •O<sub>2</sub> were generated and participated in the degradation process of PFOA.

In order to further promulgate the roles of various active species in the degradation of PFOA, the potential energy surface under the attack

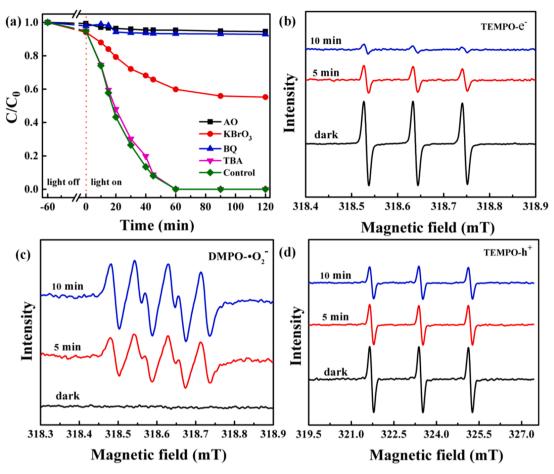


Fig. 4. (a) Trapping experiments of decomposing PFOA by add typical scavengers; EPR spectra of BOF-2 recorded for TEMPO-  $e^-$  (b), DMPO- $\bullet$ O $_2^-$  (c) and TEMPO-  $h^+$  (d) with different light irradiation time.

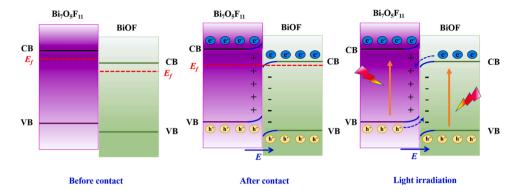
of different active species is plotted by DFT calculation. Fig. S8 presents the optimized geometries of the transition states, possible reaction pathways and the principal attack sites of PFOA under the attack of  $\bullet O_2^-$ ,  $e^-$  and  $h^+$  radicals according to our previous study [11]. The reaction barrier of  $\bullet O_2^-$ ,  $h^+$  and  $e^-$  is 24.78 kcal  $\mathrm{mol}^{-1}$ , 18.98 kcal  $\mathrm{mol}^{-1}$  and 4.56 kcal  $\mathrm{mol}^{-1}$ , respectively. Apparently, the attack of PFOA by electrons is the most thermodynamically favorable. Based on this, it is speculated that PFOA is firstly reduced by the photogenerated  $e^-$ , and then is decomposed under the joint action of  $\bullet O_2^-$  and  $h^+$ . On the other hand,  $\bullet O_2^-$  and  $h^+$  might also directly attack and decompose PFOA.

To explain the enhanced photocatalytic activity of the Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub>/ BiOF composite, the band structures of Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub> and BiOF were evaluated by UV-vis DRS, Motty-Scotty plots and DFT calculations. The Eg values of Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub> and BiOF were 3.53 and 3.30 eV, respectively. Accordingly, the ECB, EVB and Ef values of Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub> and BiOF (the detailed calculation process is presented in SI) were calculated to be -1.15, -0.47, 2.38, 2.83, -0.95 and -0.27 eV, respectively. Fig. S10 reveals the band charge density distribution in Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub>/BiOF. The highest occupied energy band (HOMO) is occupied by the [Bi<sub>2</sub>O<sub>2</sub>] layers in BiOF bulk phase, while the lowest unoccupied energy band (LUMO) is occupied by [Bi2O2] layers in the surface of BiOF and the interface region nearby Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub>. The photogenerated electrons at HOMO could be easily excited to the LUMO as the Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub>/BiOF was exposed to light. According to the band structure parameters and the migration of photogenerated carriers determined by DFT calculations, the charges in the Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub>/BiOF complex mainly concentrated in Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub> while holes reserved in BiOF. This characteristic is different from that of the traditional type II heterojunction, suggesting it is an S-scheme heterojunction photocatalyst.

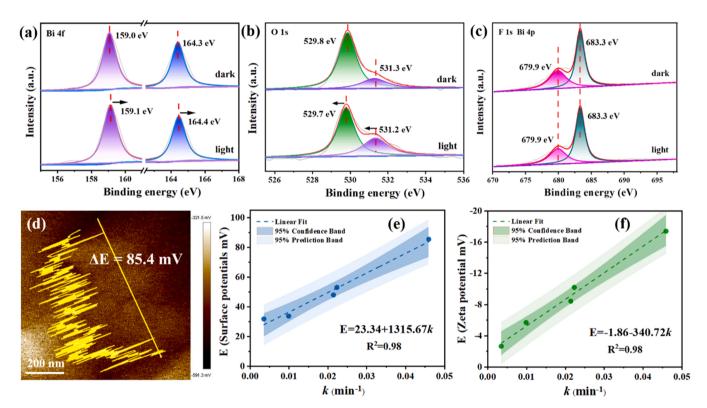
As displayed in Scheme 2,  $Bi_7O_5F_{11}$  has a higher  $E_f$  than BiOF, when  $Bi_7O_5F_{11}$  and BiOF contacted intimately, the  $e^{\cdot}$  in  $Bi_7O_5F_{11}$  transfered to BiOF until achieving a balanced  $E_f$  [24,25]. This charge transfer at the interface would induce formation of an IEF between the positively charged  $Bi_7O_5F_{11}$  and negatively charged BiOF. Furthermore, the band edge of  $Bi_7O_5F_{11}$  bends up due to loss of free electrons, while that of BiOF bends down due to enrichment of free electrons. Under light irradiation, both  $Bi_7O_5F_{11}$  and BiOF are excited, the IEF and energy band bending drive the photoproduced  $e^{\cdot}$  moving from the CB of BiOF to combine with  $h^+$  in the VB of  $Bi_7O_5F_{11}$ , thus leading to formation of an S-scheme heterojunction between  $Bi_7O_5F_{11}$  and BiOF.

In order to demonstrate the surface charge transfer pathway more accurately, in situ XPS tests were performed on BOF-2. As shown in Fig. 5, an increase in the binding energy of Bi elements and a decrease in the binding energy of O elements were observed, which may be caused by the transfer of charge from the BiOF surface to the Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub> surface under illumination. The 3D charge density difference was calculated to further probe the charge migration path and redistribution at the interface of Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub>/BiOF. Fig. S10 illustrates that charges mainly transfer from BiOF bulk phase to Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub> via the interface, resulting in the accumulation of charges in Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub>. Thus, the photogenerated electrons of BiOF bulk phase flow to Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub> surface rapidly, leaving the photogenerated holes at HOMO. The electric field direction was from the  $Bi_7O_5F_{11}$  surface to the BiOF surface, which could accelerate the transfer of electrons at the interface [40]. As a consequence, the electron mobility is improved and separation of the e<sup>-</sup>-h<sup>+</sup> pairs is facilitated, which further expounds the photocatalysis performance of Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub>/BiOF composites.

To further clarify the effect of IEF on the photoreactivity, the surface



Scheme 2. Schematic illustration of photocatalytic mechanism for BOF-2 composite under light irradiation.



**Fig. 5.** In situ XPS fine spectrum of Bi 2 f (a), O 1 s (b) and F 1 s (c) in BOF-2; Surface potentials of BOF-2 (d); the linear relationship of the surface potentials (e) and Zeta potential (f) with rate constant *k*.

potentials and Zeta potential versus photoreactivity are qualitatively correlated and shown in Fig. 5(e, f). The IEF is reported to be proportional to the square root of the product of the surface potential and the surface charge density [41,42]. Kelvin probe force microscopy (KPFM) detects the surface potential. The surface potentials ( $\Delta E$ ) of BOF-1, BOF-2, BOF-3, BOF-4 and BOF-5 are approximately 53.1 mV, 85.4 mV, 48.0 mV, 33.8 mV and 31.9 mV (Fig. 5d and Fig. S11), respectively. BOF-2 exhibits the highest surface potential among the five kinds of BOFs. The surface charge density may be proportional to the Zeta potential [42]. The measured zeta potential changes (Table S5) were as follows: BOF-2 (-17.4 mV) > BOF-1 (-10.2 mV) > BOF-3 (-8.43 mV) > BOF-4 (-5.69 mV) > BOF-5 (-2.66 mV). BOF-2 again exhibits the largest Zeta potential, fortifying its highest surface charge density. In addition, the surface potentials and Zeta potential are positively correlated with the rate constant k, verifying that the enhancement of the IEF intensity is an essential factor in facilitating the performance of BOF-2.

The separation efficiency of the photoexcited electron-hole pairs was

further explored by photoluminescence spectroscopy (PL) and transient photocurrent response (PC). As shown in Fig. 6a, the PL spectra of the asprepared  $\text{Bi}_7\text{O}_5\text{F}_{11}/\text{BiOF}$  materials were analyzed at 325 nm excitation wavelength. There was a wide emission peak between 400 and 650 nm for all the materials. The emission intensity of the samples significantly decreased following the order of BOF-5  $\approx$  BOF-4 > BOF-3 > BOF-2, but then increased. Thus, the BOF-2 exhibited the weakest PL intensity among the prepared materials. In agreement with this, the transient photocurrent response of BOF-2 was much higher than that of other materials (Fig. 6b). These demonstrated that the electron-hole recombination rate of the BOF-2 was inhibited most efficiently, thereby prolonging the life time of the photogenerated-electrons, which was beneficial to the generation of  $\bullet \text{O}_2^-$  radical and the improvement of photocatalytic performance.

## 3.4. Degradation pathway of PFOA

The calculation of mineralization rate by analyzing the removal rate

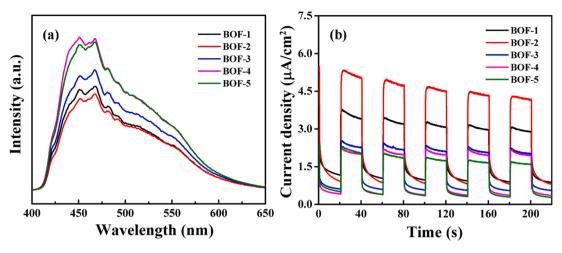


Fig. 6. PL spectra (a) and transient photocurrent response (b) of the as-prepared Bi<sub>7</sub>O<sub>5</sub>F<sub>11</sub>/BiOF materials.

of TOC in PFOA solution is of great significance to avoid secondary pollution in wastewater treatment. Fig. S12a displays that the removal rate of TOC increased with illumination time and reached 53% at 2 h. It was reported that the removal rate of TOC of PFOA by Pb-BFO/0.5% RGO system was only 28.0% after 8 h UV light irradiation [43]. In order to further analyze the specific process of PFOA decomposition, UPLC/MS/MS was used to explore the produced intermediates in the degradation process. A range of perfluorocarboxylic acids (PFCAs), including perfluorobutanoic acid (PFBA,  $C_3F_7COOH$ ; m/z = 213), peracid (PFPeA, C<sub>4</sub>F<sub>9</sub>COOH; m/z = 263), fluoropentanoic fluorohexanoic acid (PFHxA; C<sub>5</sub>F<sub>11</sub>COOH; m/z = 313) perfluoroheptanoic acid (PFHpA;  $C_6F_{13}COOH$ ; m/z = 363) were detected and quantified in the reaction solution (Fig. S13). As shown in Fig. S12b, the defluorination rate reached 29% within 2 h UV light irradiation, which proved that the C-F bonds were effectively destroyed. Song et al. [44] reported that the defluorination efficiency in PFOA degradation was 12% after 16 h of reaction with traditional Ti<sub>3</sub>C<sub>2</sub>-HF/TiO<sub>2</sub> as catalyst.

#### 4. Conclusions

 $Bi_7O_5F_{11}$  and BiOF have similar layered structures but different band structures, which were ingeniously used to fabricate a high-performance  $Bi_7O_5F_{11}/BiOF$  photocatalyst. The hydrolysis of Bi(III) ions together with the coordination between Bi(III) ions and  $NH_4F$  were regulated to control the local  $BiF_3$  accumulation in the composite. When the mass ratio of  $Bi_7O_5F_{11}$  and BiOF was 3:1, the corresponding BOF-2 exhibited the most outstanding photocatalytic performance to complete degrade PFOA. This work further revealed that the introduction of a "build-in electric field" into the "S-scheme" photocatalyst has a strong driving force to enhance the performance of the composite photocatalyst. The reaction might be initiated by the reaction between PFOA and the highly reductive e generated by the irradiated BOF-2, then decomposed by  $h^+, \bullet O_2^-$ . Attacked by these active species, PFOA was gradually mineralized by removing CF2 units.

#### CRediT authorship contribution statement

Jingzhen Wang: Conceptualization, Investigation, Resources, Data curation, Writing – original draft. Chun-Shuai Cao: Conceptualization, Resources, Data curation. Yinqing Zhang: Supervision, Writing – review & editing. Lingyan Zhu: Conceptualization, Supervision, Writing – review & editing, Supervision, Project administration, Funding acquisition.

## **Declaration of Competing Interest**

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

#### **Data Availability**

Data will be made available on request.

#### Acknowledgements

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## Appendix A. Supporting information

Supplementary data associated with this article can be found in the online version at doi:10.1016/j.apcatb.2023.122899.

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